

Fig. 4. Coordination polyhedra of $\operatorname{Sm}(9)$ and $\operatorname{Sm}(10)$. Assumed sphere radii are proportional to atomic radii.

In spite of the complexity of the structure which presents 16 site sets for the Pd atoms, they are four, five or six coordinated with Sm atoms in only a few ways. Around $\mathrm{Pd}(7), \mathrm{Pd}(8)$ and $\mathrm{Pd}(11)$, the Sm atoms form a slightly puckered square; $\operatorname{Pd}(1), \operatorname{Pd}(2), \operatorname{Pd}(3), \operatorname{Pd}(4)$, $\operatorname{Pd}(5), \operatorname{Pd}(6)$ and $\operatorname{Pd}(10)$ are at the centre of a more or less irregular trigonal bipyramid; the environment of $\operatorname{Pd}(12), \operatorname{Pd}(13), \operatorname{Pd}(15)$ and $\operatorname{Pd}(16)$ appears to approximately tetragonal pyramidal but with the axial $\mathrm{Sm}-\mathrm{Pd}$ bond generally longer than the equatorial ones; $\operatorname{Pd}(9)$ is octahedrally surrounded; finally, $\operatorname{Pd}(14)$ is a hybrid, as it is coordinated with four Sm and one $\mathrm{Pd}(14)$, the latter giving the only Pd-Pd distance which shows a slight contraction ( $2.71 \AA$ ).

Finally, we compare the present structure with that of the previously mentioned cubic Laves phase, which is adopted by Eu and by the alkaline earths $\mathrm{Ca}, \mathrm{Sr}$ and Ba (Pearson, 1967) in the corresponding $M \mathrm{Pd}_{2}$ compounds. Although these two structural types do have something in common (e.g. both are layered), they are quite different. In $\mathrm{Sm}_{10} \mathrm{Pd}_{21}$ short contacts occur only between atoms of a different kind. Moreover, whereas in the Laves phase Pd is surrounded by an icosahedron, i.e. a bicapped pentagonal antiprism, in $\mathrm{Sm}_{10} \mathrm{Pd}_{21}$, even if all the atoms up to the second gap are considered, the environments around $\operatorname{Pd}(12)$,
$\operatorname{Pd}(13), \operatorname{Pd}(15)$ and $\operatorname{Pd}(16)$ can be described as bicapped pentagonal prisms, and the situation differs further for the remaining Pd atoms.

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# Synthesis and Crystal Structure of Diamagnetic ReP $_{4}$, a Polyphosphide with Re-Re Pairs 

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#### Abstract

Microcrystalline $\mathrm{ReP}_{4}$ was prepared by heating the elemental components in the presence of iodine. Single crystals were obtained by reaction of the components


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in molten tin. They are orthorhombic, Pbca, $a=$ 6.227 (2), $b=9.231$ (2), $c=10.854$ (3) $\AA, Z=8$. The structure is of a new type. It was determined and refined from single-crystal diffractometer data to $R=$ 0.042 for 1195 structure factors. The Re atoms have
six P neighbors forming a distorted octahedron. The P atoms are approximately tetrahedrally coordinated by Re and P atoms. In common with other transitionmetal polyphosphides with this composition, all nearneighbor interactions can be rationalized as twoelectron bonds. Thus the Re atoms obtain formal oxidation numbers +2 ( $d^{5}$ system). They are displaced from the centers of their P octahedra to permit $\mathrm{Re}-\mathrm{Re}$ bonding ( $3.012 \AA$ ) via edges of the paired octahedra. In this way all spins are compensated and this is confirmed by the diamagnetism of the compound. The $\mathbf{P}$ atoms form puckered ten-membered rings which are condensed to a two-dimensionally infinite net, which is a variation of the net found in the $\mathrm{CrP}_{4}$ structure.

## Introduction

In the first systematic study of the rheniumphosphorus system Haraldsen (1935) identified four phases with the approximate compositions $\operatorname{Re}_{2} \mathrm{P}, \operatorname{ReP}$, $\mathrm{ReP}_{2}$, and $\mathrm{ReP}_{3}$. Later on, the structures of the two compounds with relatively high Re content were determined from powder data and their ideal compositions were established as $\mathrm{Re}_{2} \mathrm{P}$ (Rundqvist, 1961) and $\mathrm{Re}_{3} \mathrm{P}_{4}$ (Rundqvist, 1966). The compounds with high phosphorus content remained poorly characterized, apparently because of the difficulty in growing crystals of sufficient size for structure determinations. In the course of our reinvestigations of the phosphorusrich parts of transition-metal-phosphorus systems, employing previously little used preparative methods, we have already reported on $\mathrm{FeP}_{4}$ (Jeitschko \& Braun, 1978), and on $\mathrm{RuP}_{4}$ and $\mathrm{OsP}_{4}$ (Braun \& Jeitschko, 1978a). We have now grown crystals of $\mathrm{ReP}_{4}$ and determined their structure.

## Synthesis, crystal growth and properties

In agreement with the aforementioned experiences, we have not obtained single crystals of rhenium polyphosphides by direct reaction of the elemental components. Apparently, these negative results are due to the inertness of red P at moderate ( $<800 \mathrm{~K}$ ) temperatures (Schäfer \& Binnewies, 1978), whereas at higher temperatures the decomposition pressure of the polyphosphide may be exceeded. However, we have readily prepared micröcrystalline Re polyphosphides by adding iodine to the mixture of the elemental components. Using tin as a reaction medium, we have obtained welldeveloped single crystals.

Starting materials were powders of $\operatorname{Re}$ (stated purity $99.99 \%$ ), iodine (Merck, reinst), $\operatorname{Sn}$ ( $99.9 \%$ ), and red P (Merck, rein), which was treated with an aqueous solution of NaOH (Brauer, 1975) to dissolve oxidation
products and dried in a desiccator with $\mathrm{P}_{2} \mathrm{O}_{5}$. Samples with compositions varying between $\mathrm{Re}: \mathrm{P}=1: 4$ to 1:20 were heated in evacuated silica tubes with about 1 to $5 \mathrm{at} . \%$ iodine at temperatures of about 1200 K for 7 d . For the preparation with the tin flux the atomic ratios $\mathrm{Re}: \mathrm{P}: \mathrm{Sn}$ varied between 1:5:9 and 1:30:28 with annealing for 7 d at about 1050 K . The matrix of tin and tin phosphides was dissolved in moderately dilute ( $1: 1$ ) hydrochloric acid. In no case did the powder photographs of the resultant products show any indication of a Re phosphide with a phosphorus content higher than $\mathrm{ReP}_{4}$. We have, however, obtained single crystals of a $\operatorname{Re}$ phosphide with a somewhat lower P content, which we are investigating.*

A sample of $\mathrm{ReP}_{4}$ prepared in the tin flux was analyzed by X-ray fluorescence spectroscopy. No elements other than Re and P were found. The detectability limit for tin was about $1 \mathrm{wt} \%$; however, as judged from previous experiences (Jeitschko \& Braun, 1977), the solid solubility of such polyphosphides for tin should be much lower than that.

The single crystals of $\mathrm{ReP}_{4}$ used for the structure determination were taken from a sample prepared in an evacuated alumina container with initial composition $\operatorname{Re}: \mathrm{P}: \mathrm{Sn}=1: 30: 28$, heated within a few hours to 1400 K , slowly cooled to 1100 K within 7 d , and cooled to room temperature in 5 min .

The $\mathrm{ReP}_{4}$ crystals are shiny black with welldeveloped faces and no pronounced growth directions. They are stable in air and not attacked by nonoxidizing acids. Magnetic measurements with the Faraday technique indicate $\mathrm{ReP}_{4}$ to be diamagnetic.

## Cell dimensions and space group

Single crystals of $\mathrm{ReP}_{4}$ were investigated in Weissenberg and Buerger precession cameras. They showed orthorhombic symmetry. Space-group extinctions (reflections $0 k l$ were observed only with $k=2 n, h 0 l$ only with $l=2 n$, and $h k 0$ only with $h=2 n$ ) led to the unique space group $\operatorname{Pbca}\left(D_{2 h}^{15}\right)$.

The lattice constants were refined from Guinier powder data with $a$-quartz ( $a=4.9130, c=5.4046 \AA$ ) as standard: $a=6.227(2), b=9.231(2), c=$ 10.854 (3) $\AA, V=623.9$ (4) $\AA^{3}$. Assuming $Z=8$ formula units per cell, the calculated density is 6.60 Mg $\mathrm{m}^{-3}$. Samples prepared by the different techniques and different heat treatments had lattice constants which were within the error limits given above in parentheses for the least significant digits. This suggests that the compound has only a narrow homogeneity range and that any inclusion of iodine or tin must be very limited.

[^0]Table 1. Powder diagram of $\mathrm{ReP}_{4}$
The experimental powder pattern was obtained in a Guinier camera with $\mathrm{Cu} K \alpha$ radiation. For the calculated pattern (Yvon, Jeitschko \& Parthé, 1977), data of the refined structure were used. The $Q$ values are defined by $Q=10^{4} / d^{2}\left(A^{-2}\right)$.

| nkl | ${ }^{\circ} \mathrm{c}$ | $Q_{0}$ | ${ }^{1}{ }_{c}$ | ${ }^{1}$ | hkl | $Q_{c}$ | - | ${ }^{1}{ }_{c}$ | Io | hkl | $Q_{c}$ | - | ${ }^{\text {I }}$ | 1 。 |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 002 | 340 | 340 | 21 | $s$ | 230 | 2097 | 2088 | 14 | - | 116 | 3432 | - |  |  |
| 111 | 460 | 462 | 71 | vvs | 231 | 2172 | - | 0 | - | 135 | 3436 | - | 2 | - |
| 020 | 469 | 471 | 2 | vo | 042 | 2217 | 2217 | 10 | $s$ | 234 | 3446 | 3446 | 8 | m |
| 021 | 554 | 555 | 16 | $s$ | 141 | 2220 | - | 1 | - | 331 | 3461 | 3463 | 5 | $\cdots$ |
| 102 | 597 | 599 | 24 | vo | 223 | 2265 |  | 0 | - | 144 | 3494 | 3494 |  | vs |
| 112 | 715 | 716 | 14 | $s$ | 204 | 2390 | - | 0 | - | 026 | 3526 \} | 526 | 1) |  |
| 022 | 809 | - | 1 |  | 232 | 2427 | 2428 | 26 | va | 152 | 3531 \} | 3526 | 1) |  |
| 121 | 812 | 813 | 46 | vo | 142 | 2475 | 2475 | 17 | s | 323 | 3554 | - | 1 |  |
| 200 | 1031 | - | 0 | - | 115 | 2498 | 2497 | 7 | m | 225 | 3623 | - | 0 | - |
| 122 | 1067 | - | 0 |  | 214 | 2507 |  | 1 | - | 243 | 3673 | 3674 | ${ }^{8}$ | va |
| 113 | 1139 | - | 1 | - | 311 | 2523 | 2523 | 14 | 5 | 304 | 3679 |  | $12\}$ | vo |
| 210 | 1149 | 1150 | 31 | s | 025 | 2592 | 2591 | 10 | m | 332 | 3716 | 3716 | 5 | * |
| 211 | 1234 \} | 1234 | 22 | vo | 043 | 2642 | 2641 | 4 | * | 126 | 3784 | - | 0 | - |
| 023 | 1234 \} |  | 100 |  | 302 | 2660 | 2660 | 14 | 8 | 314 | 3796 | 3796 | 12 | $s$ |
| 004 | 1358 | - | 0 |  | 134 | 2672 | 2673 | 29 | vs | 153 | 3956 | - | 0 | - |
| 202 | 1371 | - | 0 | - | 312 | 2777 | 2777 | 9 | m | 250 | 3965 | 3964 | 2 | $n$ |
| 131 | 1399 | 1400 | 3 | $\omega$ | 125 | 2850 ${ }^{\text {3 }}$ | 2851 | 9 ) |  | 045 | 4000 | 4000 | 1 | vv* |
| 212 | 1488 | 1488 | 7 | w | 233 | 2852 ${ }^{\text {2 }}$ \} | 2851 | 16 | v | 251 | 4050 | 4049 | 9 | \% |
| 123 | 1491 | - | 1 | - | 224 | 2859 | - | 1 | - | 206 | 4088 | - | 0 | - |
| 220 | 1501 | 1502 | 13 | s | 321 | 2875 | 2875 | 23 | vs | 400 | 4125 | 4125 | 3 |  |
| 221 | 1586 | - | 0 | - | 143 | 2900 | - | 0 | - | 333 | 4141 | - | 0 |  |
| 104 | 1616 | 1616 | 7 | m | 240 | 2909 |  | 0 | - | 324 | 4148 | 4148 | 3 |  |
| 132 | 1654 | 1655 | 27 | ve | 241 | 2994 |  | 0 | - | 216 | 4205 | 4206 | 13 | s |
| 114 | 1734 | 1734 |  | * | 006 | 3056 | 3056 | 26 | v8 | 235 | 4210 | - | 1 | - |
| 024 | 1828 | 1827 | 1 | vvw | 322 | 3129 | 3131 | 2 | vw | 060 | 4225 | - | 4 | - |
| 222 | 1840 | - | 0 | - | 313 | 3202 | - | 0 | - | 410 | 4242 | - | 1 | - |
| 040 213 | 1878 | 1878 | ${ }^{6}$ | \% | 044 | 3236 | 3235 | 3 | $\underline{\square}$ | 145 | 4258 | - | $\bigcirc$ |  |
| 213 | 1963 | 1913 | 30 1 | va | 242 | 3249 3271 |  |  | - | 344 | ${ }_{4}^{4287}$ | 4283 |  |  |
| 133 | 2078 | 1963 | 0 | vw | 151 | 3277 ${ }^{3}$ | 3273 | 23 , | vo | 252 | 4305 | 428 | 0 | - |
| 124 | 2086 | - | 0 | - | 106 | 3314 | 3314 |  | vw |  |  |  |  |  |

The powder diagrams of $\operatorname{ReP}_{4}$ (Table 1) have only a vague resemblance to the diagram drawn by Haraldsen (1935) for the Re phosphide with the highest $P$ content found by him and to which he ascribed the tentative composition $\mathrm{ReP}_{3}$.

## Structure determination and refinement

A total of 1897 unique reflections were measured on a four-circle diffractometer with graphite-monochromated Mo Ka radiation, a scintillation counter and pulse-height discriminator. $\omega$ scans were taken with a speed of $0.05^{\circ} \mathrm{s}^{-1}$ and a scan angle of $1^{\circ}$. Background was counted for 22.5 s at both ends of the scan. All reflections within one octant up to $2 \theta=80^{\circ}$ were measured. The crystal had well-developed faces, but for the purpose of the absorption correction - was approximated by a sphere of radius 0.037 mm . The linear absorption coefficient $\mu_{\text {Mo } K \alpha}$ is $39.0 \mathrm{~mm}^{-1}$.

The locations of the Re atoms were deduced from a Patterson map and the positions of the P atoms were

Table 2. Atom parameters of $\mathrm{ReP}_{4}$
All atoms are in the general position of space group Pbca. Positional parameters are multiplied by $10^{4}$. Numbers in parentheses are e.s.d.'s in the least significant digits. The last column contains $B$ values ( $\AA^{2}$ ) as obtained in a least-squares refinement with isotropic thermal parameters.

|  | $x$ | $y$ | $z$ | $B$ |
| :--- | :---: | :---: | :---: | :---: |
| Re | $1218(1)$ | $1095(1)$ | $0755(1)$ | $0.53(1)$ |
| $\mathrm{P}(1)$ | $0613(5)$ | $2829(3)$ | $2364(3)$ | $0.70(4)$ |
| $\mathrm{P}(2)$ | $1889(5)$ | $9479(3)$ | $2386(3)$ | $0.60(3)$ |
| $\mathrm{P}(3)$ | $7538(5)$ | $0436(3)$ | $0817(3)$ | $0.64(3)$ |
| $\mathrm{P}(4)$ | $4986(5)$ | $2089(3)$ | $0693(2)$ | $0.62(3)$ |

obtained from difference Fourier syntheses. The structure was refined with a full-matrix least-squares program (Sheldrick, 1976) using scattering factors for neutral atoms (Cromer \& Mann, 1968), corrected for anomalous dispersion (Cromer \& Liberman, 1970). Weights were assigned according to counting statistics. An isotropic extinction parameter was refined and applied to the $F_{c}$ values. Reflections which were overcorrected by this procedure, as well as reflections with $F_{o}<6 \sigma$ were assigned zero weight in the last leastsquares cycles. For a refinement with isotropic thermal parameters a conventional $R$ value of 0.043 was obtained for the 1195 reflections with non-zero weight. The introduction of anisotropic thermal parameters reduced $R$ only to 0.042 . Thus the anisotropic parameters deviate only slightly from the isotropic ones.*

[^1]Table 3. Interatomic distances ( $\AA$ ) in $\mathrm{ReP}_{4}$
Standard deviations are all less than $0.005 \AA$. All distances shorter than $3.6 \AA$ (for Re) and $3.1 \AA$ (for P atoms) are listed.


Fig. 1. Stereoplot of the $\mathrm{ReP}_{4}$ structure drawn with Johnson's (1965) program. Filled circles: Re, open circles: P.


Fig. 2. Projection of the $\operatorname{ReP}_{4}$ structure along the $x$ axis. The left hand side shows the sheets of edge- and corner-shared ReP ${ }_{6}$ octahedra extending parallel to the $x y$ plane. The two-dimensional P poly-'anions' extending parallel to the same plane are emphasized in the right hand part of the drawing. Coordinates in the $x$ direction are given in hundredths.

For the total of 1897 unique reflections, $R$ equals 0.086 . A difference synthesis revealed no features lower than -3.6 e $\AA^{-3}$ or higher than +3.1 e $\AA^{-3}$. Atomic parameters and interatomic distances are given in Tables 2 and 3. The structure is shown in Figs. 1 and 2.

## Discussion

The structure of $\mathrm{ReP}_{4}$ is of a new type and is best discussed together with the other polypnictides of that stoichiometry which are summarized in Table 4. In these ten compounds - which crystallize in seven different structure types - the metal atoms are always approximately octahedrally coordinated by pnicogen atoms ( $\mathrm{P}, \mathrm{As}$ ) and the pnicogen atoms are all tetrahedrally coordinated, one half of them by two pnicogen and two metal atoms and the other half by one metal and three pnicogen atoms. If classical two-electron bonds are assumed for each of these near-neighbor interactions and if the electrons in the metal-pnicogen bonds are counted as belonging to the pricogen atoms, the metal atoms all obtain formal oxidation numbers

Table 4. Pnictides with composition $M \mathrm{P}_{4}$ and $\mathrm{MAs}_{4}$
Isostructural compounds are enframed. The numbers of $d$ electrons not used for $M-\mathrm{P}$ bonding are indicated. References are (a) Krebs, Müller \& Zürn (1956); (b) Jeitschko \& Donohue (1972); (c) Jeitschko \& Donohue (1975); (d) Gibinski, Cisowska, Zdanowicz, Henkie \& Wojakowski (1974); El Maslout, Zanne, Jeannot \& Gleitzer (1975); von Schnering \& Menge (1976); (e) Gerardin, Aubry, Courtois \& Protas (1977); (f) Braun \& Jeitschko (1978a); (g) Jeitschko \& Braun (1978); $(h)$ this work.

| $\mathrm{CrP}_{4}{ }^{\text {b }}$ | MnP4 ${ }^{\text {c }}$ | $\mathrm{FeP}_{4}{ }^{\text {g }}$ |  |
| :---: | :---: | :---: | :---: |
| MoP4 ${ }^{\text {b }}$ |  | $\mathrm{RuP}_{4}{ }^{\text { }}$ |  |
|  | $\mathrm{ReP}_{4}{ }^{\text {n }}$ | OsP4 ${ }^{\prime}$ | $\mathrm{MgAs}_{4}{ }^{\text {® }}$ |
| $d^{4}$ | $d^{5}$ | $d^{8}$ | $d^{0}$ |

+2. Thus, for instance, Cr in $\mathrm{CrP}_{4}$ obtains a $d^{4}$ system and Re in $\mathrm{ReP}_{4}$ a $d^{5}$ system.

Considering their tetrahedral environment, $s p^{3}$ hybridization can be assumed for the pnicogen atoms. In the essentially ionic Mg and Cd compounds, the electrons of the metal-pnicogen interactions will certainly belong predominantly to the pnicogen atoms and thus may also be considered as lone-pair electrons of the pnicogen anions which are directed towards the metal cations. In the transition-metal pnictides, these electrons will essentially be covalent and interacting with the $d^{2} s p^{3}$ hybrid of the metal atoms. Since the $e_{g}$ ( $d_{x^{2}-y^{2}}$ and $d_{z}{ }^{2}$ ) orbitals are used for this interaction, the remaining valence electrons of the metal atoms have to be accommodated in the $t_{28}\left(d_{x y}, d_{x z}\right.$ and $\left.d_{y z}\right)$ orbitals of the metal atoms which therefore all obtain low-spin systems. Thus, in $\mathrm{FeP}_{4}$ the Fe atoms obtain a low-spin $d^{6}$ system and the compound is diamagnetic. In analogy, low-spin $d^{6}$ systems were assumed for the metal atoms in $\mathrm{RuP}_{4}$ and $\mathrm{OsP}_{4}$ and recent magnetic measurements (Braun \& Jeitschko 1978b) confirm these compounds to be diamagnetic.

Like Mn in $\mathrm{MnP}_{4}, \operatorname{Re}$ in $\mathrm{ReP}_{4}$ has a $d^{5}$ system which permits metal-metal bonding by interaction of one of the half-occupied $t_{2 g}$ orbitals of each metal atom. As can be seen in Fig. 3, the $\mathrm{ReP}_{6}$ octahedra are paired by sharing one edge. The distortions of the octahedra clearly indicate the bonding $\mathrm{Re}-\mathrm{Re}$ interaction: the tetrahedral bonding angles of ideally $109^{\circ}$ are compressed to $79^{\circ}$ and the octahedral angle of ideally $90^{\circ}$ is stretched to $101^{\circ}$ to permit a close approach of the two Re atoms. The diamagnetism of $\mathrm{ReP}_{4}$ proves that the spins are compensated in a covalent bond. The $\mathrm{Re}-\mathrm{Re}$ distance of $3.012 \AA$ is within the range $2.96-$ $3.04 \AA$ found for unbridged single Re-Re bonds in organometallic compounds (Ciani, Sironi \& Albano, 1977).

The average $\mathrm{Re}-\mathrm{P}$ distance of $2.402 \AA$ compares favorably with the average $\mathrm{Ru}-\mathrm{P}$ distance of $2.367 \AA$ in $\mathrm{RuP}_{4}$ since atomic, covalent and ionic radii of Ru are all about $0.03 \AA$ shorter than the corresponding radii of Re . It is also within the range of $\mathrm{Re}-\mathrm{P}$ distances


Fig. 3. Near-neighbor environments of the $\operatorname{Re}$ pairs in $\operatorname{ReP}_{4}$. Distances are given in $\dot{A}$, angles in degrees of arc. Standard deviations are all less than $0.005 \AA$ and $0.15^{\circ}$.
observed in molecular Re phosphine compounds with octahedral coordination of the Re atoms. Thus, for instance, for the two mutually trans $\mathrm{Re}-\mathrm{P}$ interactions in $\mathrm{ReH}_{2}(\mathrm{NO})\left(\mathrm{PPh}_{3}\right)_{3}$, the $\mathrm{Re}-\mathrm{P}$ distance is $2.39 \AA$ (Ciani, Giusto, Manassero \& Albinati, 1976). In $\left(\mathrm{PPh}_{3}\right)_{2}(\mathrm{CO})_{2} \mathrm{Re}\left(\mathrm{O}_{2} \mathrm{CCH}_{3}\right)$ the average $\mathrm{Re}-\mathrm{P}$ distance is $2.415 \AA$ (La Monica, Cenini, Forni, Manassero \& Albano, 1976); in $\left(\mathrm{Ph}_{3} \mathrm{P}\right)_{2}(\mathrm{CO})_{2} \operatorname{Re}\left(\mathrm{~S}_{2} \mathrm{CH}\right)$ it is 2.418 $\AA$ (Albano, Bellon \& Ciani, 1971); in $\operatorname{ReCl}\left(\mathrm{N}_{2}\right)\left(\mathrm{PMe}_{2} \mathrm{Ph}\right)_{4}$ it equals $2.422 \AA$ (Davis \& Ibers, 1971). In [( $\left.\left.\mathrm{Me}_{3} \mathrm{P}\right)_{2}(\mathrm{CO})_{3} \mathrm{Re}\right]_{2} \mathrm{Se}$, the $\mathrm{Re}-\mathrm{P}$ distances vary between 2.46 and $2.47 \AA$ (Röttinger, Küllmer \& Vahrenkamp, 1978) and in ( $\mathrm{PhMe}_{2} \mathrm{P}_{4} \mathrm{ClReN}_{2} \mathrm{MoCl}_{4}{ }^{-}$ (OMe) they range from 2.471 to $2.486 \AA$ (Mercer, 1974). With a range of individual $\mathrm{Re}-\mathrm{P}$ bond lengths from 2.347 to $2.521 \AA$, the variation in $\operatorname{ReP}_{4}$ is even greater than in the aforementioned molecular compounds.

The short $\mathrm{Re}-\mathrm{P}$ distances in $\mathrm{ReP}_{4}$ occur in general on that side of the $\mathrm{ReP}_{6}$ octahedron which is adjacent to the neighboring edge-shared $\operatorname{ReP}_{6}$ octahedron. This is in an analogous way also true for $\mathrm{MnP}_{4}$ and can be rationalized by the $\operatorname{Re}-\mathrm{Re}(\mathrm{Mn}-\mathrm{Mn})$ bonding interaction. If one assumes some rigidity of the $\mathrm{P}_{6}$ octahedron due to $\mathrm{P}-\mathrm{P}$ repulsion, the displacements of the Re atoms in the two edge-shared $\operatorname{ReP}_{6}$ octahedra


Fig. 4. Projections of partial structures of $\mathrm{ReP}_{4}$ and $\mathrm{CrP}_{4}$ along axes perpendicular to the two-dimensional networks of the phosphorus atoms. The linking of the phosphorus atoms is shown on the left hand sides. At the right hand sides, the metal atoms above and beneath the phosphorus poly-'anion' are shown as large filled and open circles, respectively. The linkage of the $M \mathrm{P}_{6}$ octahedra is shown in each drawing only for one metal atom layer. The phosphorus octahedra are somewhat idealized to simplify the projections.
toward each other will shorten the distances of the Re atoms to the bridging P atoms, and lengthen the distances to the P atoms at the opposite side.

The $\mathrm{Re}-\mathrm{P}-\mathrm{P}$ bond angles vary from 104.3 to $131.6^{\circ}$ with an average of $116 \cdot 3^{\circ}$. The $\mathrm{P}-\mathrm{P}-\mathrm{P}$ bond angles cover the range from 88.4 to $109.4^{\circ}$ with the average value of $97.5^{\circ}$. Thus the average $\mathrm{Re}-\mathrm{P}-\mathrm{P}$ angle is greater than the average $\mathrm{P}-\mathrm{P}-\mathrm{P}$ angle. This can be rationalized with size considerations, since the Re atoms are larger than the P atoms. The $\mathrm{Re}-\mathrm{P}(4)-\mathrm{Re}$ angle is $124.3^{\circ}$. Those $\mathrm{P}-\mathrm{Re}-\mathrm{P}$ angles which are $90^{\circ}$ in a regular octahedron vary between 83.7 and $96.1^{\circ}$ with the exception of the one angle of $101^{\circ}$ already discussed above.

The P atoms in $\mathrm{ReP}_{4}$ form ten-membered rings which are found with the same conformation in many other $M \mathrm{P}_{4}$ compounds. These rings are condensed to two-dimensionally infinite nets which are related to the nets found in $\mathrm{CrP}_{4}$ (Fig. 4). In $\mathrm{ReP}_{4}$, adjacent rows of rings extending along the $x$ direction are antiparallel to each other while every other such row is parallel (zigzag arrangement). In $\mathrm{CrP}_{4}$ all rows are parallel (zig-zig arrangement). The different linking of these tenmembered rings is also reflected in the linking of the $M \mathrm{P}_{6}$ octahedra which form edge-shared pairs or infinite chains respectively. More complicated variations of these building principles seem possible. Thus, for instance, a zig-zig-zag sequence of the rows would correspond to chains of four edge-shared octahedra followed by pairs of edge-shared octahedra. In the structures of $\mathrm{FeP}_{4}, \mathrm{RuP}_{4}$, and $\mathrm{CdP}_{4}$, the tenmembered rings of P atoms are also condensed to infinite two-dimensional nets; however, the linking is different (Jeitschko \& Braun, 1978).

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# The Crystal Structure of Tetraaqua-cis-dichloroiron(III) Tetrachloroferrate(III) Monohydrate (Ferric Chloride 2 $\frac{1}{2}$-Hydrate)* 

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#### Abstract

The crystal structure of ferric chloride $2 \frac{1}{2}$ hydrate, $\mathrm{FeCl}_{3} .2 \frac{1}{2} \mathrm{H}_{2} \mathrm{O}, \quad\left[\mathrm{FeCl}_{2}\left(\mathrm{H}_{2} \mathrm{O}\right)_{4}\right]^{+} .\left[\mathrm{FeCl}_{4}\right]^{-} . \mathrm{H}_{2} \mathrm{O}$, has been determined from single-crystal X -ray diffractometer data using an average of three equivalent data sets, and refined to $R=0.035$ for the 1617 unique reflections collected using Mo $K a$ radiation. The structure is orthorhombic, space group $P b c 2_{1}$, with $a=$ 6.272 (3), $b=12.945$ (3) and $c=16 \cdot 654$ (4) $\AA, Z=4$, and consists of a somewhat distorted tetrahedral $\mathrm{FeCl}_{4}^{-}$ anion, a distorted octahedral tetraaqua-cis-dichloroiron(III) cation and a single solvate water molecule. The tetrahedral $\mathrm{Fe}-\mathrm{Cl}$ bonds range from $2 \cdot 169$ (4) to 2.220 (4) $\AA$. The octahedral $\mathrm{Fe}-\mathrm{Cl}$ bonds are 2.236 (4) and 2.251 (4) $\AA$, and the Fe -water distances range from 2.004 (8) to $2 \cdot 116$ (7) $\AA$. An extensive network of hydrogen bonds links the ions and the solvate water molecule.


[^2]
## Introduction

The ferric chloride-water phase diagram (Roozeboom, 1892) reveals four hydrates: $\mathrm{FeCl}_{3} .2 \mathrm{H}_{2} \mathrm{O}$, $\mathrm{FeCl}_{3} \cdot 2 \frac{1}{2} \mathrm{H}_{2} \mathrm{O}, \mathrm{FeCl}_{3} \cdot 3 \frac{1}{2} \mathrm{H}_{2} \mathrm{O}$ and $\mathrm{FeCl}_{3} \cdot 6 \mathrm{H}_{2} \mathrm{O}$. The last of these has been examined crystallographically by Lind (1967), and shown to consist of tetraaqua-transdichloroiron(III) ions, with a $\mathrm{Cl}^{-}$ion and two solvate water molecules. The crystal structure of ferric chloride $2 \frac{1}{2}$ hydrate was determined to provide a known structural background for the interpretation and use of parameters determined from infrared, Mössbauer and ultraviolet-visible absorption spectra of both the solid compound and the acidic ferric chloride solutions used to leach sulfide ores. Concentration and structural information about the complexes in these solutions is a necessary first step in the study of reaction mechanisms of these leaching reactions. This paper reports the results of the first of three crystal structure determinations of the three lower ferric chloride hydrates.


[^0]:    * Note added in proof: The structure determination of this compound resulted in the composition $\mathrm{Re}_{6} \mathrm{P}_{13}$. A detailed report will be published shortly.

[^1]:    * Lists of anisotropic thermal parameters and structure factors have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 34403 (13 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CHI 2HU, England.

[^2]:    * Minerals Research Program, Processing Contribution No. 74.

